

Assessment of High Chromium Oxide Dispersion Strengthened Steels for Nuclear Applications

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Contents

- Introduction and Objectives
- Experimental procedures
 - ODS steels
 - Hydrogen charging and thermally aged conditions
- Results
 - Hydrogen effects
 - Transition of fracture strain, changes of fracture mode
 - Critical hydrogen concentration
 - Thermal Aging effects
 - SP ductile to brittle transition behavior
 - Hardening by thermal aging
- Summary

Introduction

- Oxide dispersion strengthening (ODS) steels
 - Candidate material for fusion reactor blanket, fuel clad in FBR, and high burn-up LWR

Requirements for fuel clad application

- high resistance to irradiation embrittlement and swelling
- corrosion resistance in high temperature water environment
- low susceptibility to hydrogen-induced cracking and SCC
- lower DBTT shift by thermal aging embrittlement

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Corrosion properties

- Strong corrosion resistance in SCWR

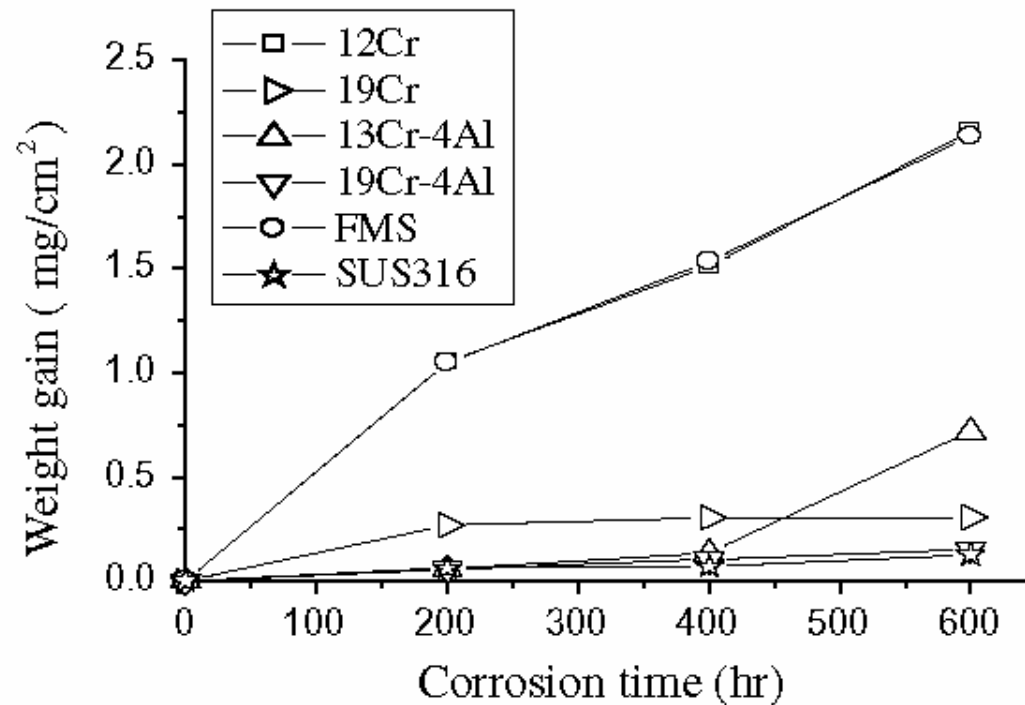


Fig. 2. The effect of Cr and Al on the corrosion properties of ODS ferritic/martensitic steel in SCPW (783 K, 25 MPa).

- General hydrogen and thermal aging effects in a high Cr steel
 - reduction in ductility and toughness
 - brittle fracture (intergranular or transgranular)
- Objectives
 - to investigate the *effects of hydrogen and thermal aging on the mechanical properties* of ODS steels depending on the anisotropy of deformation structure

Experimental

■ Test materials

-K1; 18.37Cr-0.29W-0.28Ti-0.368Y₂O₃

-K2; 13.64Cr-1.65W-4.12Al-0.28Ti-0.381Y₂O₃

-K3; 16Cr-1.82W-4.59Al-0.28Ti-0.368Y₂O₃

-K4; 18.85Cr-1.83W-0.28Ti-4.61Al-0.368Y₂O₃

-K5; 22.05Cr-1.8W-4.55Al-0.27Ti-0.356Y₂O₃

-RMS; 9Cr-2W-0.03Al

Fe-xCr-yW-zAl / Argon
gas atomized powder

Y₂O₃, Ti
Powders

Mechanically alloyed by ball mill

Canning and degassing

Hot extrusion

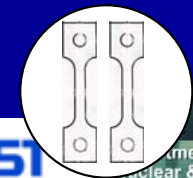
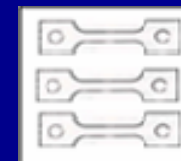
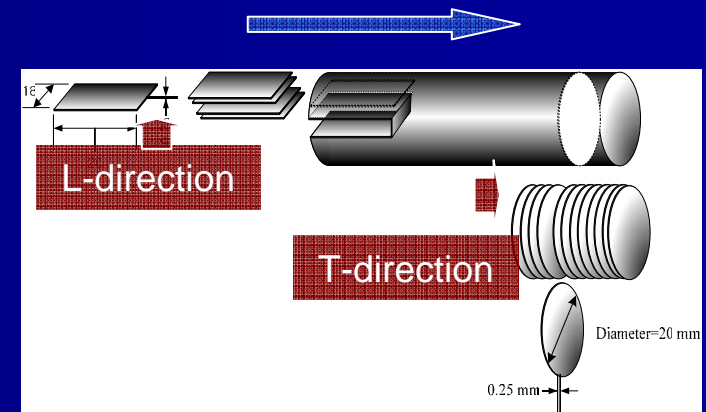
(1423 K, 67 mm dia. to 25 mm dia.)

Heat treatment

(1423K x 1h+ air cooling)

Machining

Extrusion direction



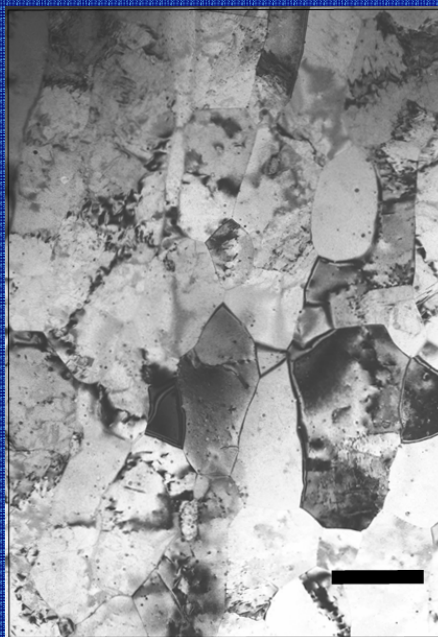
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Quantum Engineering

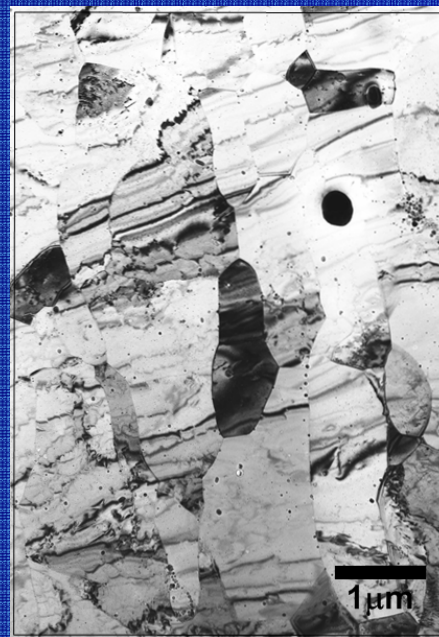
Microstructural characteristics of ODS Steels



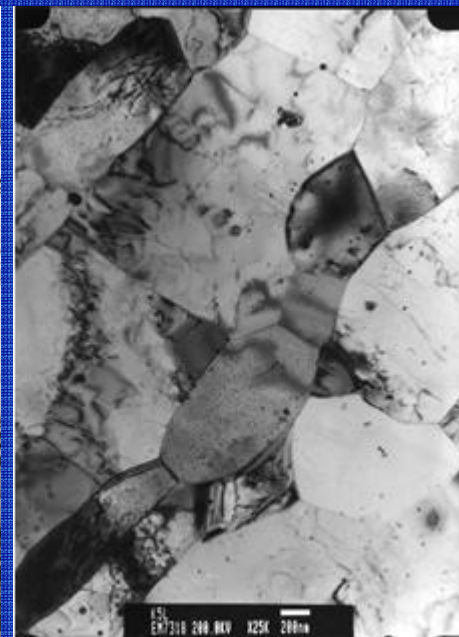
K1 ODS



K2 ODS



K4 ODS



K5 ODS

Size & number density of oxide particles

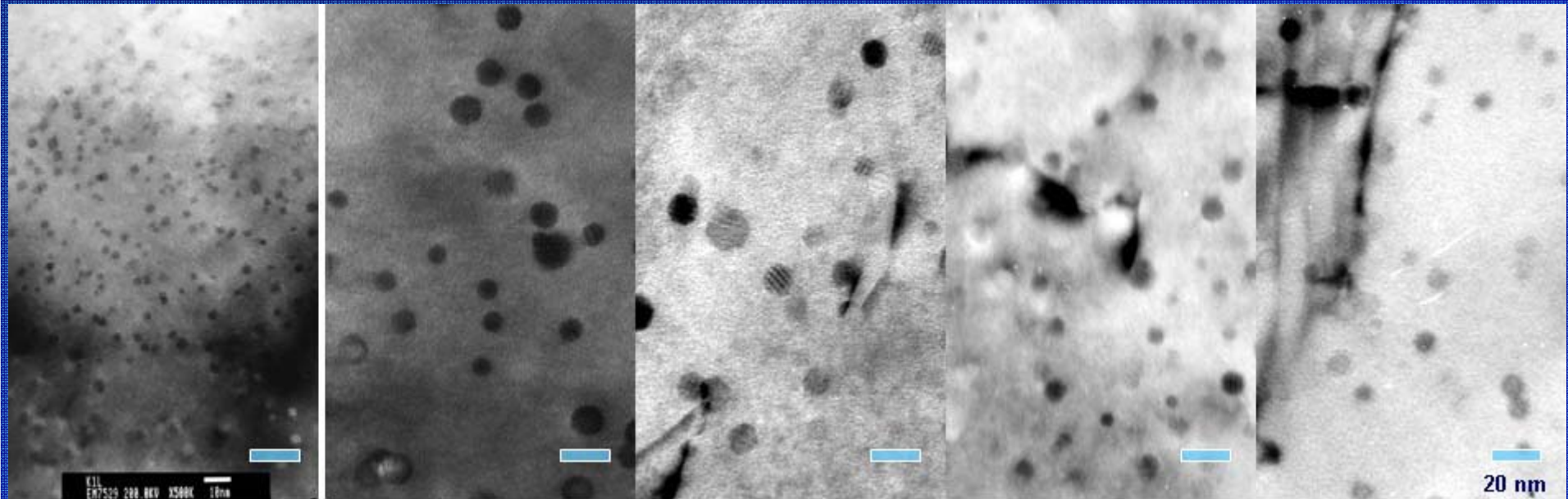
K1 ODS

K2

K3

K4

K5



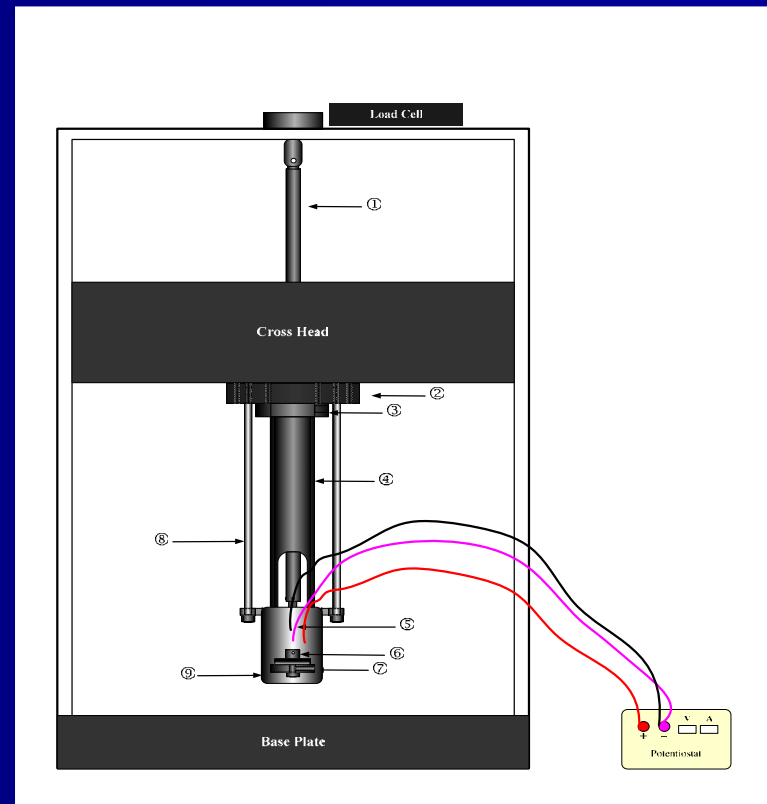
	Size (nm)	Density (m^{-3})	Average spacing (nm)
K1 (19Cr)	2.4	1.4×10^{23}	65
K2 (13Cr-4Al)	6.3	2.3×10^{22}	95
K3 (16Cr-4Al)	6.8	1.4×10^{22}	118
K4 (19Cr-4Al)	6.7	1.6×10^{22}	112
K5 (22Cr-4Al)	7.3	1.3×10^{22}	149

Hydrogen Effects

Hydrogen charging during tensile test

- Hydrogen charging conditions
 - cathodic charging
 - solution; 1N H₂SO₄ + 10 mg/l As₂O₃
 - **pre- (30 min.) + *in-situ*** charging
 - current density; 100 ~ -520 A/m²
- Test Temperature
 - 297 K
- Strain rate
 - 9x10⁻⁵ /s
- Hydrogen contents
 - measured by TDS

Tensile Testing system

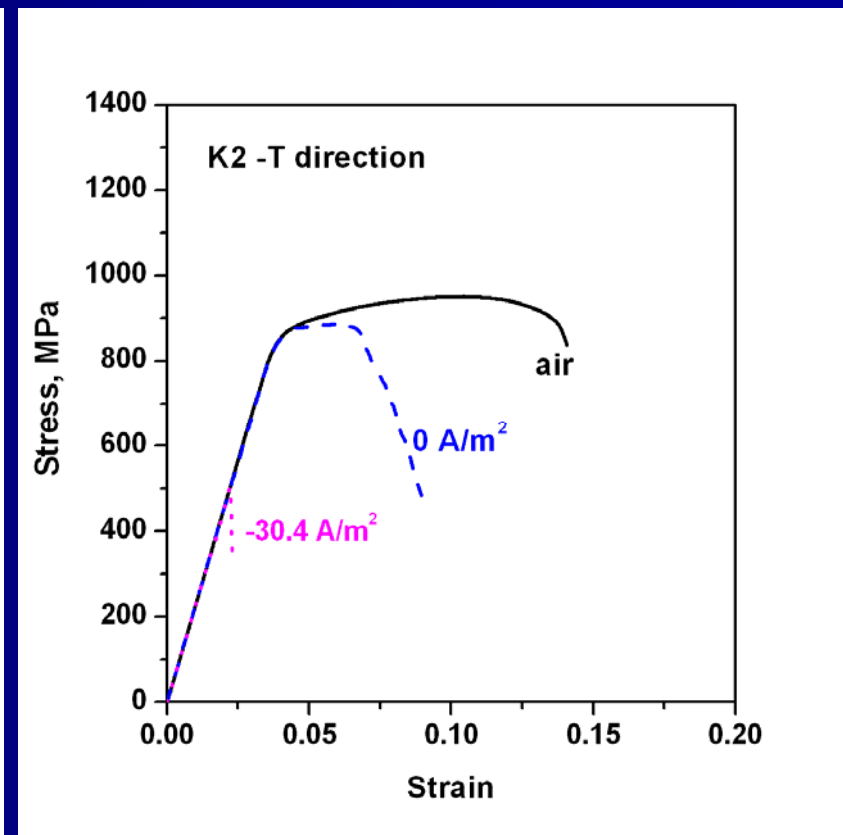
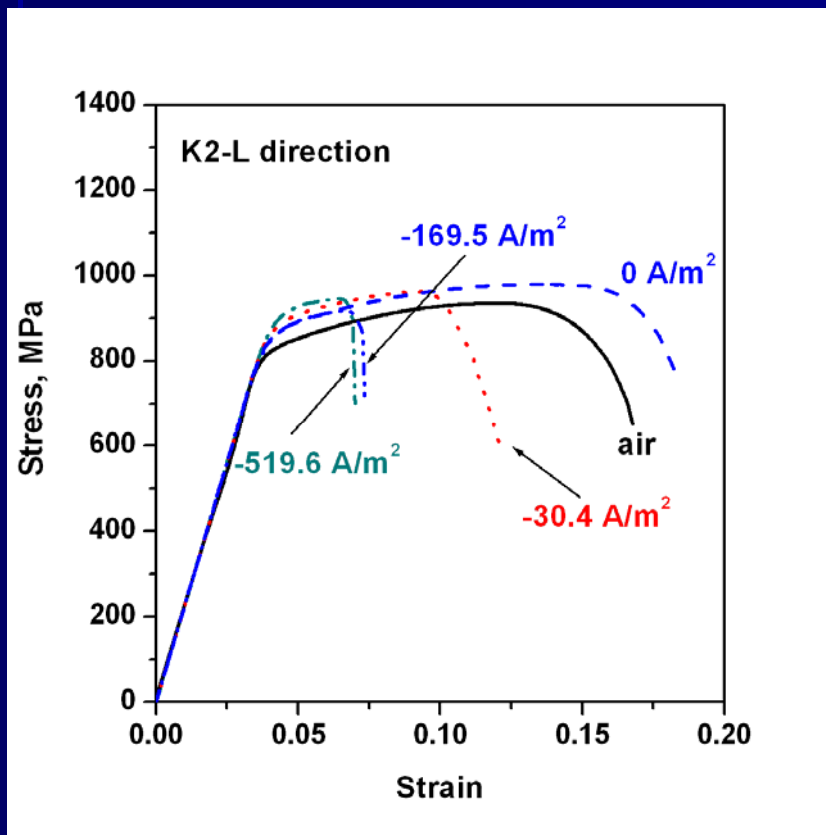


Tensile stress-strain curves

K2 ODS (13Cr-4Al)

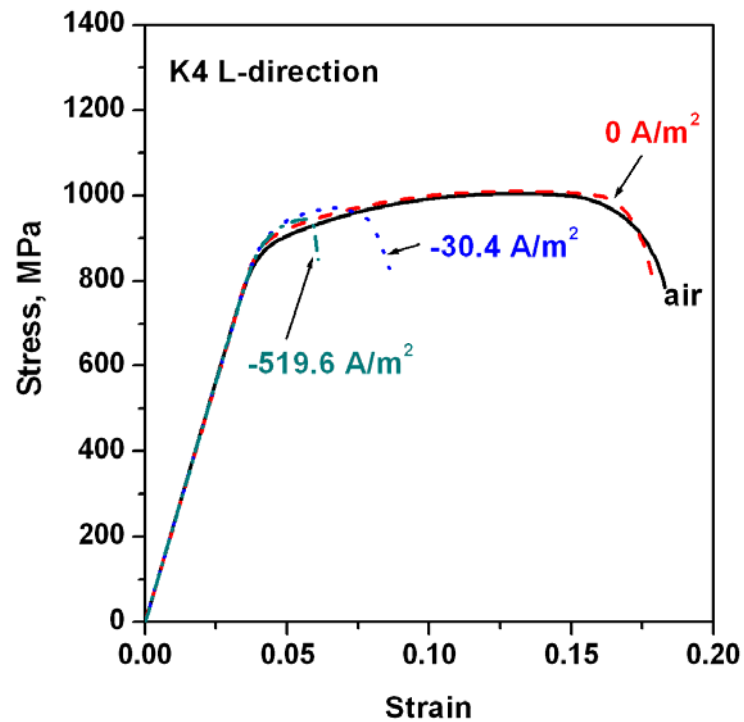
L-direction

T-direction

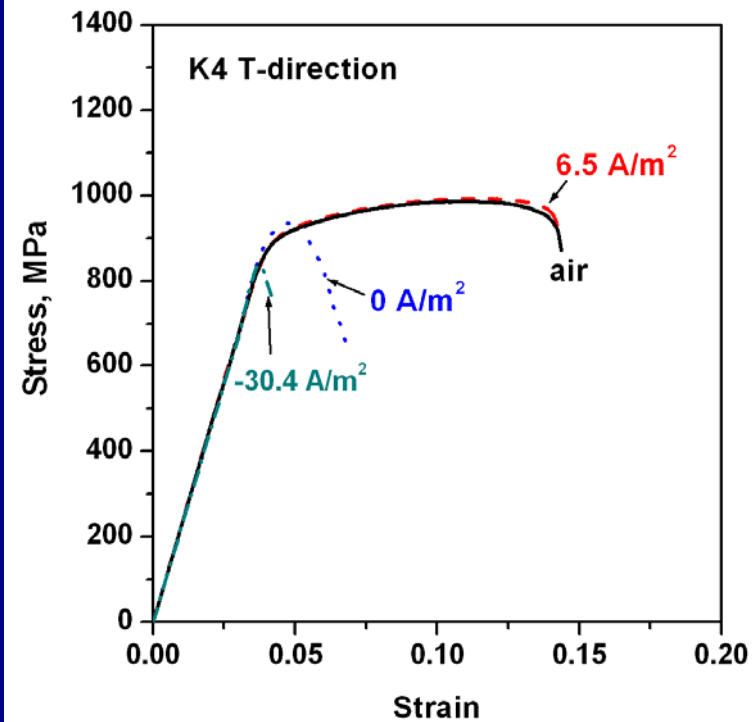


K4 ODS (19Cr-4Al)

L-direction

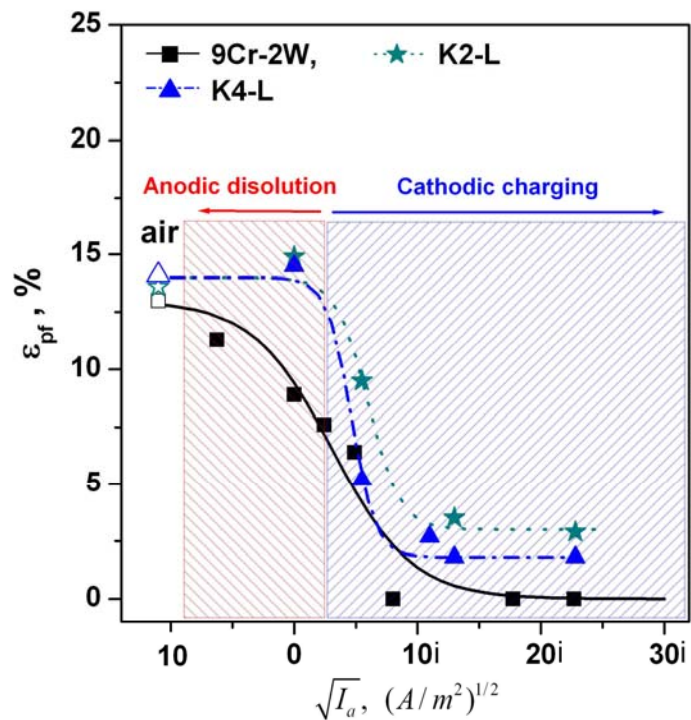


T-direction

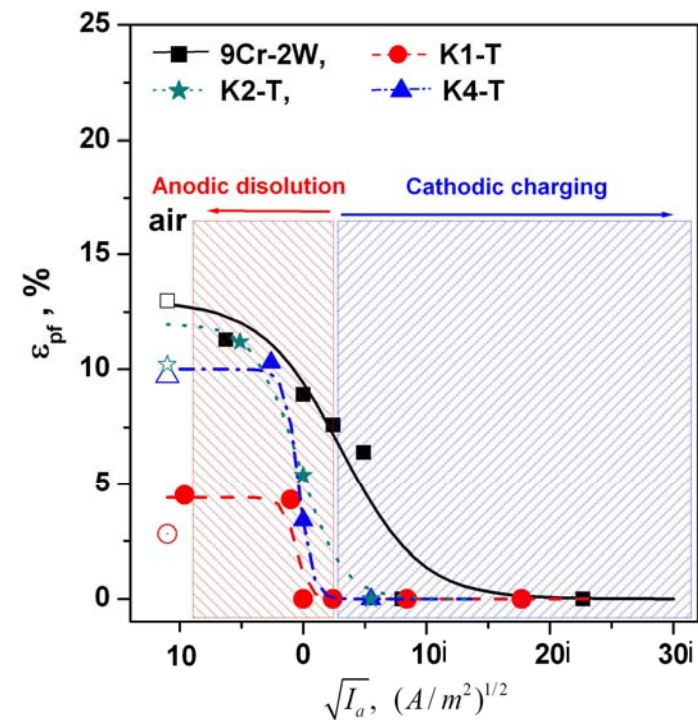


Transition of fracture strain (ϵ_{pf}) by hydrogen charging

L-direction



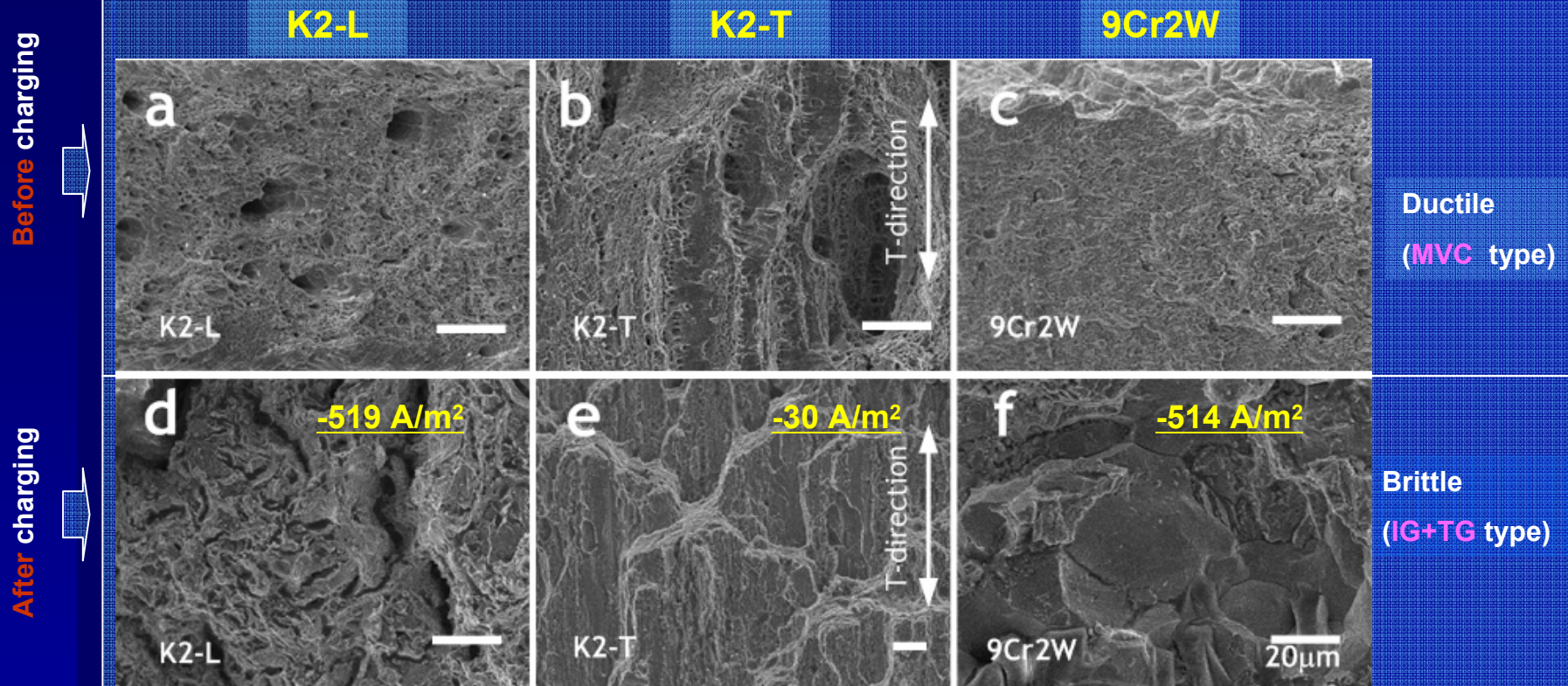
T-direction



The critical current density of K2-L, K4-L and a 9Cr2W was approximately -42.5 , -23.0 and $-4.0 A/m^2$, respectively.

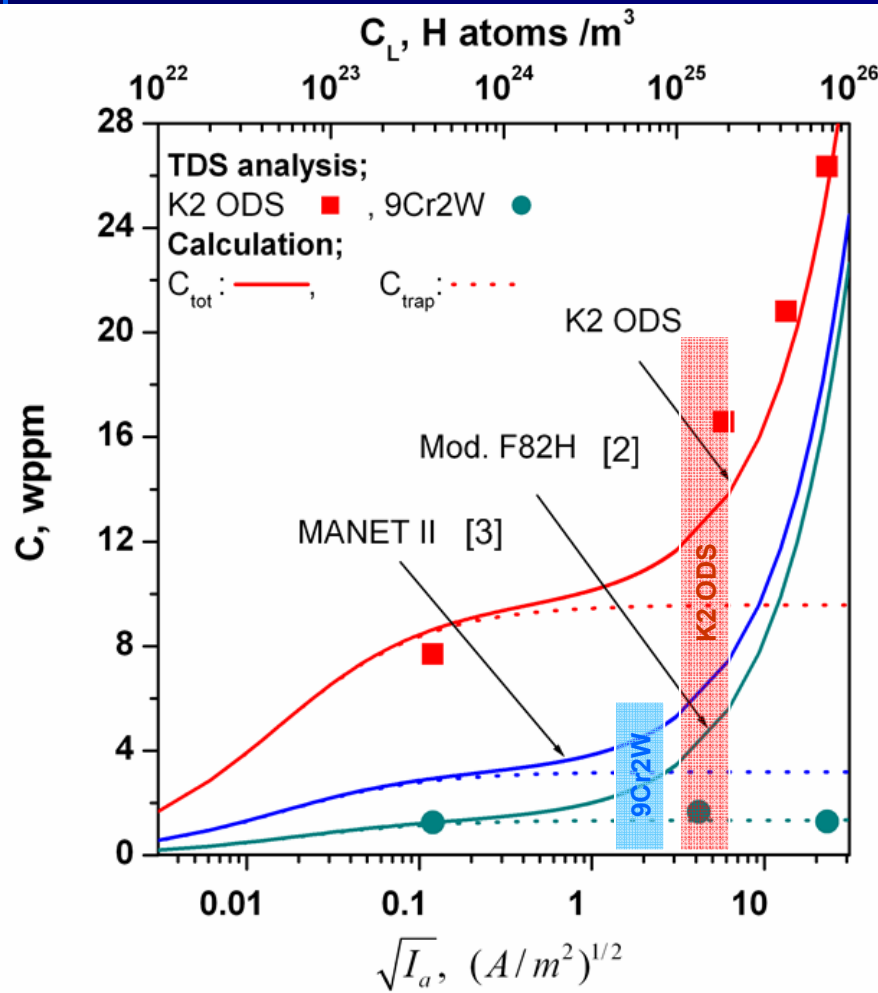
The critical current density of T-direction was almost $0 A/m^2$, irrespective of the materials.

Changes of fracture mode



Change in fracture mode from **ductile shear rupture** to brittle **inter + transgranular fracture** by H charging

Critical hydrogen concentration



H concentration in NISL and input current density in Fe iron [5].

$$C_L (\text{H atoms} / \text{gFe}) = 1.3106 \times 10^{17} \sqrt{I_m (\text{mA} / \text{cm}^2)} + 4.9645 \times 10^{16}$$

Critical hydrogen concentration of K2 ODS and 9Cr-2W steel is in the range of 10~12 and 1~2 wppm, respectively.

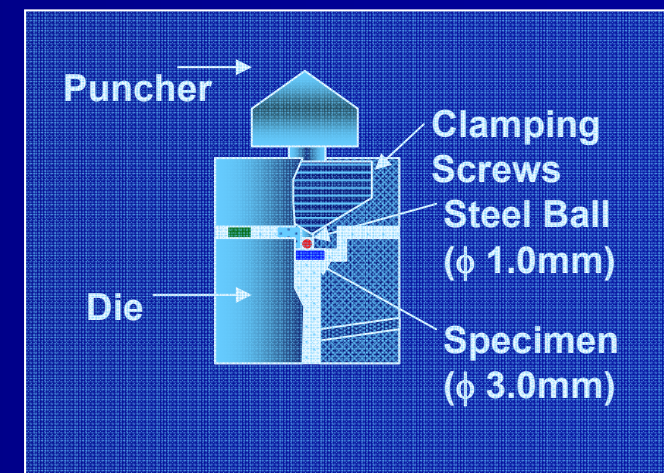
[5] J.L Lee, J.Y. Lee, Scripta METALL. 19 (1985) 341.

Thermal Aging Effects

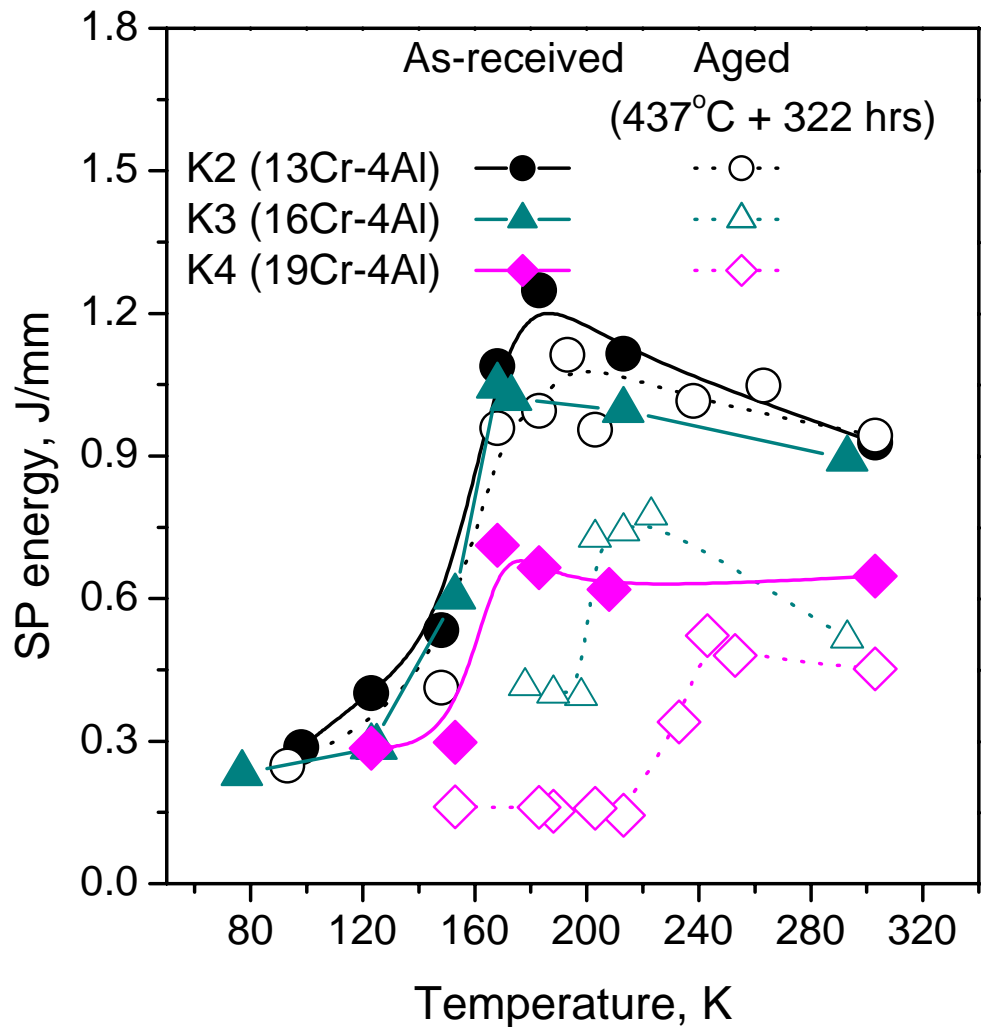
Thermal aging treatment and SP tests

- Thermal aging treatment
 - from 430 to 475 C up to 1000 h
- Small Punch test
 - TEM disk type ($\phi=3$, $t=0.25\sim 0.27$)
 - Cross head Speed: 0.2 mm/min
 - Test Temperature: from RT to 77K
 - 1 mm diameter steel ball

SP jig



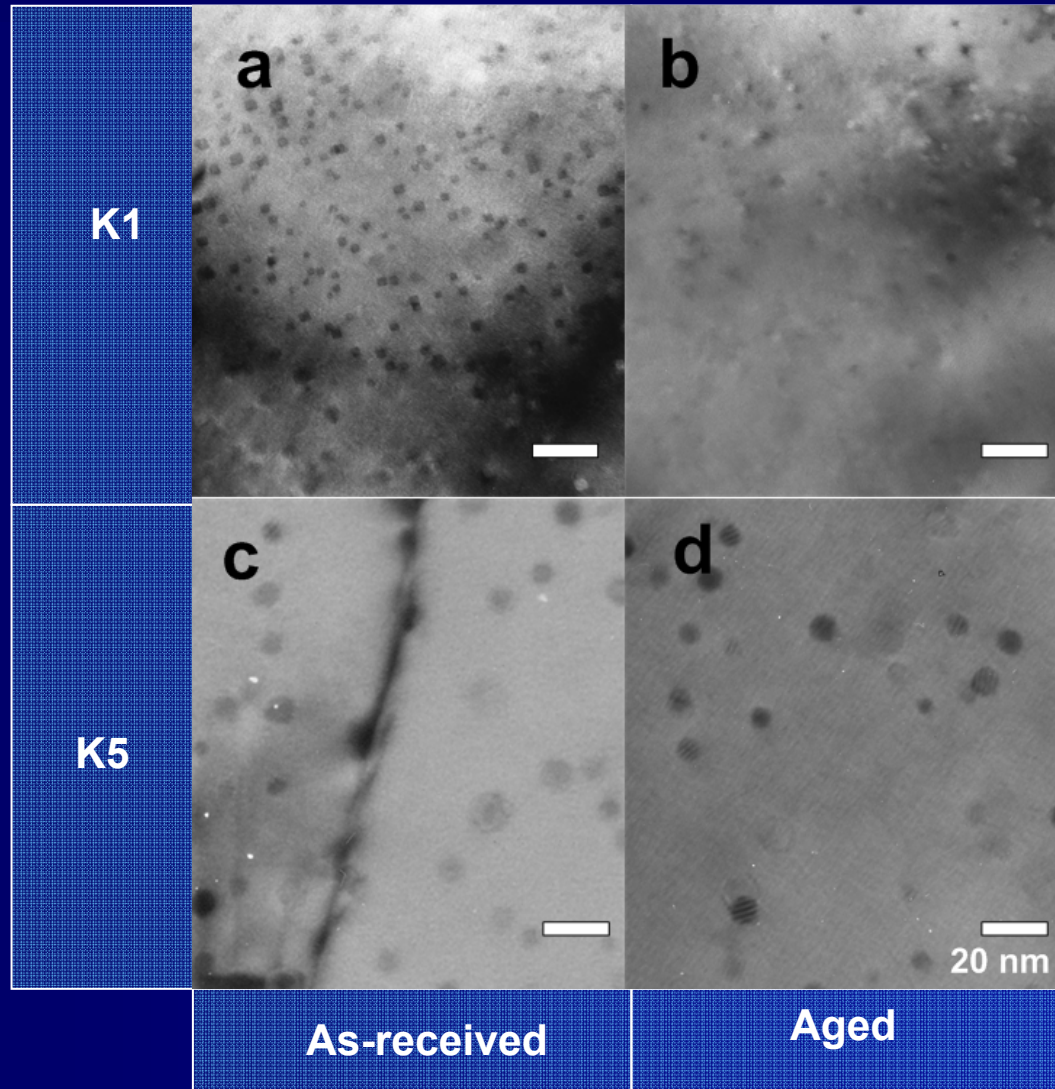
Changes in SP-energy (437 C + 322 hrs)



SP- Δ DBTT

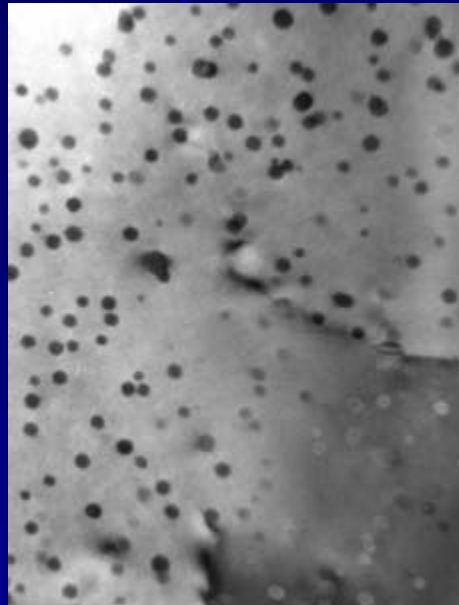
13 Cr (K2)	~10K
16Cr (K3)	~50 K
19Cr (K4)	~73 K

Thermally aged ODS steels (437 C + 322 hrs)

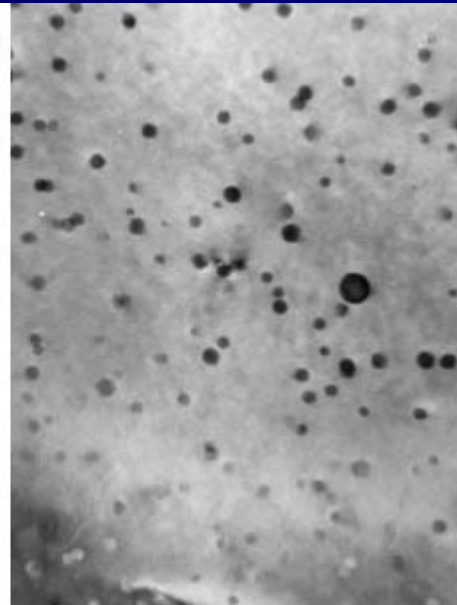


Formation of α'

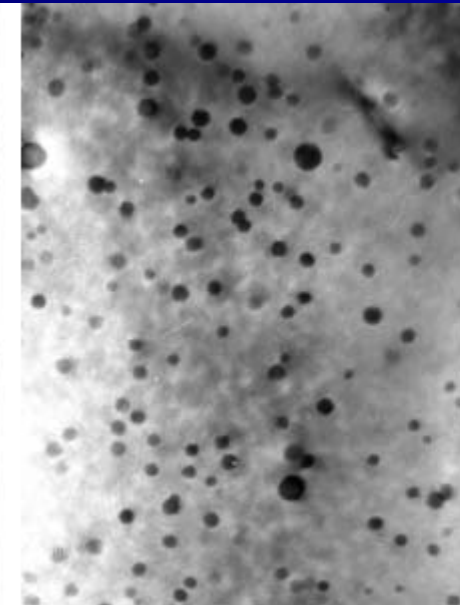
475 C Aging (K5 ODS)



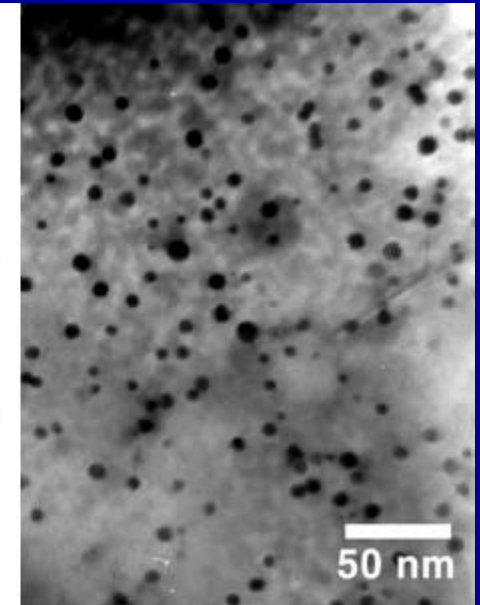
10 hours



50 hours



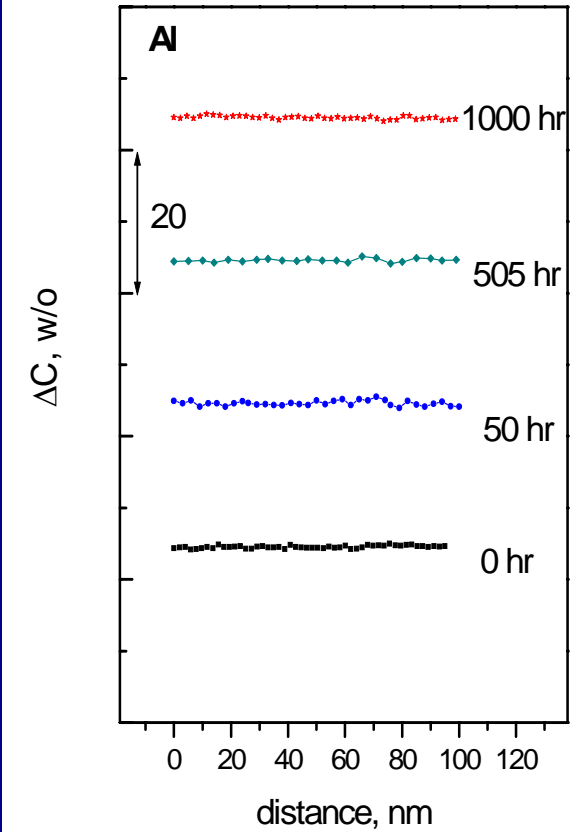
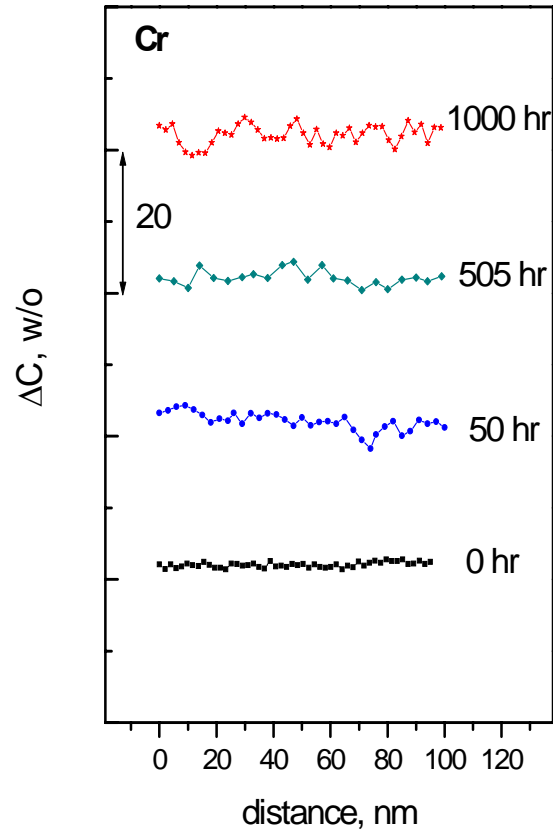
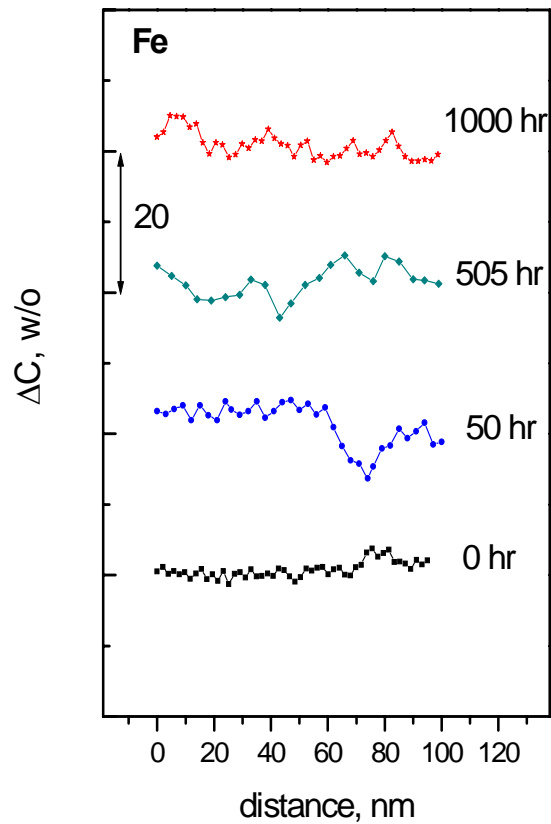
505 hours



1000 hours

Atomic fluctuation by thermal aging

475 C Aging (K5 ODS)



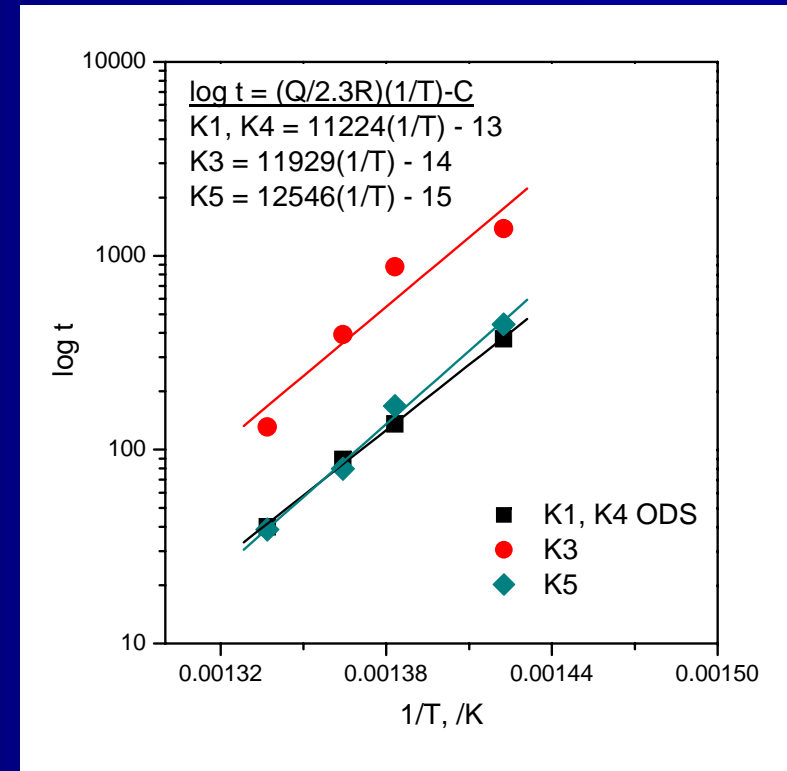
Activation energy

■ Arrhenius type

$$\begin{aligned} Q/R &= 25843 \text{ (K1, K4 ODS)} \\ &= 27468 \text{ (K3 ODS)} \\ &= 28889 \text{ (K5 ODS)} \end{aligned}$$

Activation energy

$$\begin{aligned} Q &= 214 \text{ kJ/mol (K1, K4 ODS)} \\ &= 228 \text{ kJ/mol (K3 ODS)} \\ &= 240 \text{ kJ/mol (K5 ODS)} \end{aligned}$$



*The activation energy for diffusion of Fe and Cr in a Fe-26Cr alloys was 211 and 203 kJ/mol, respectively [1].

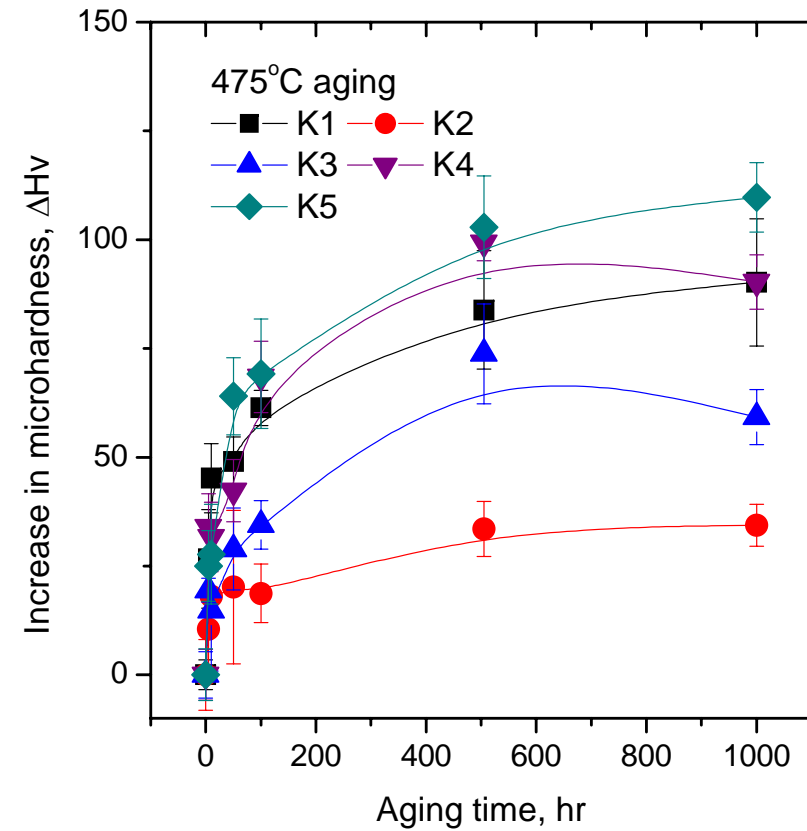
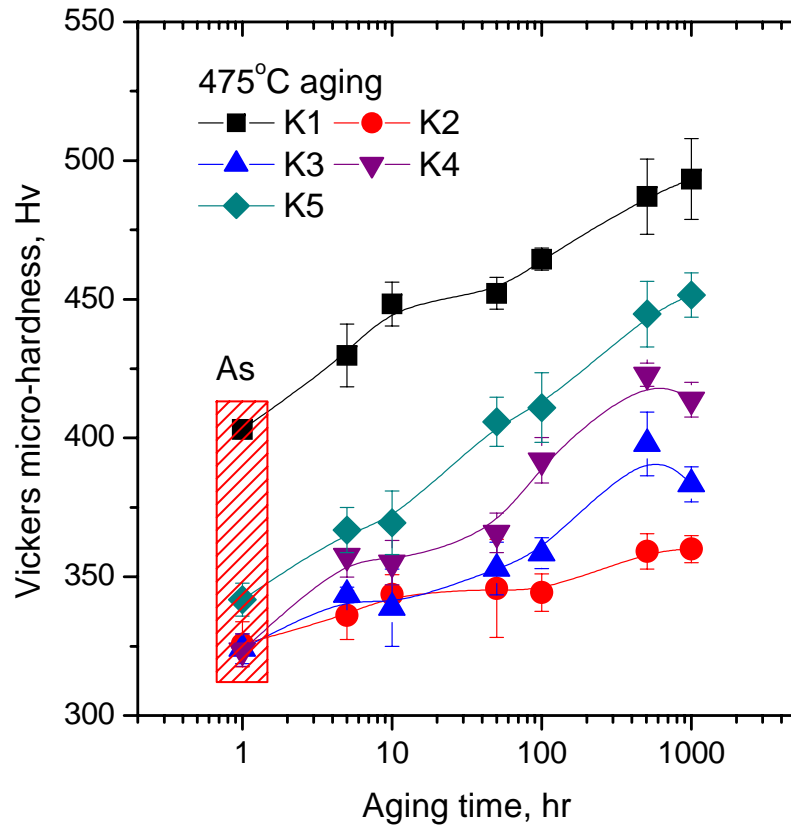
The activation energy for Cr diffusion was 229.8 kJ/mol [3]

[1] Browdes EA, Brook GB, editors. Smitells metals book. Seventh ed. Heinemann: Butterworth; 1992, p. 13

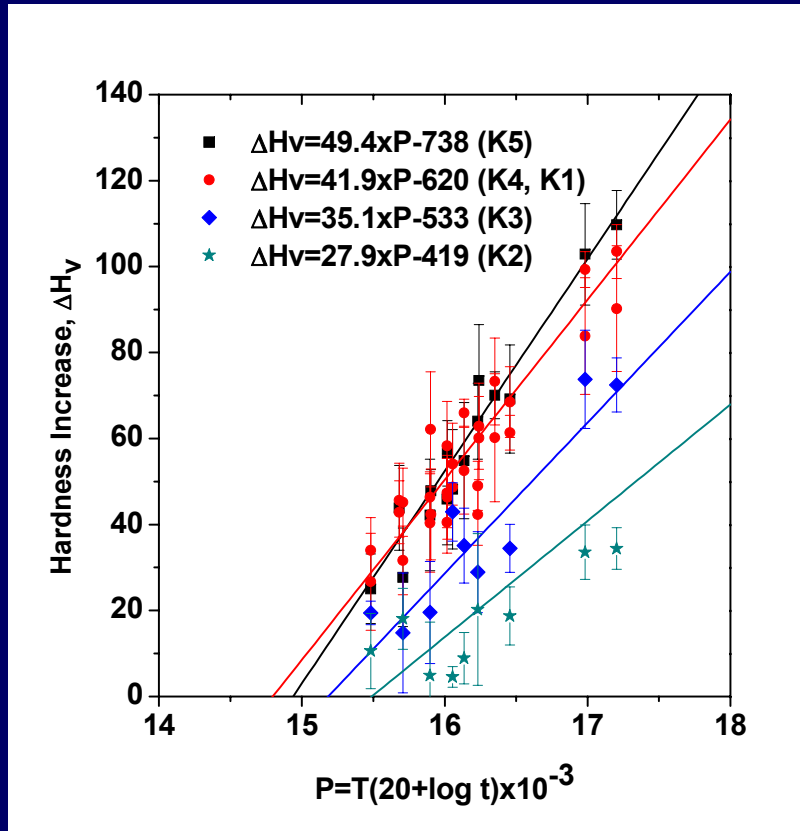
[3] T. Heumann and H. Böhmer, Arch. Eisenhüttenw., 1960, Vol. 31, p. 749

Increase in Micro-hardness

475 C Aging



Generalized Hardening Formula



$$\Delta H_v = 49.4 \times P - 738 \text{ (K5 ODS)}$$

$$\Delta H_v = 41.9 \times P - 620 \text{ (K1, K4)}$$

$$\Delta H_v = 35.1 \times P - 533 \text{ (K3)}$$

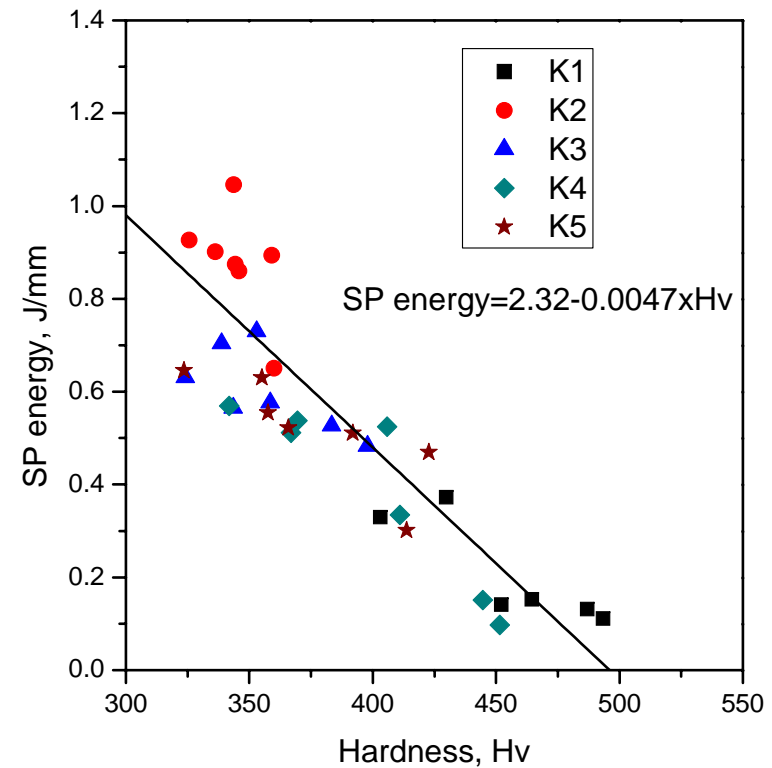
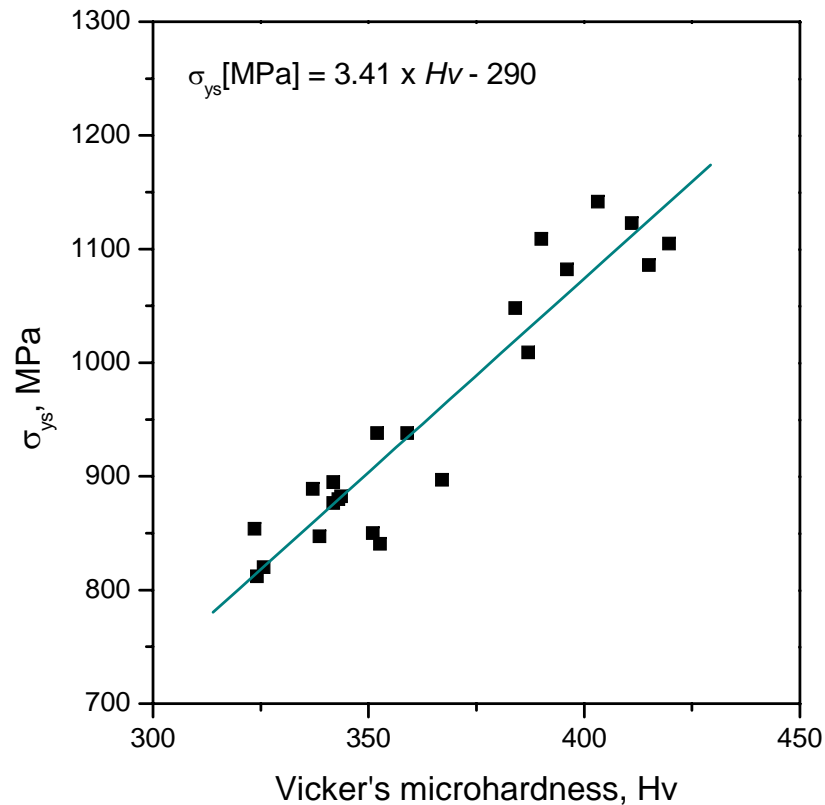
$$\Delta H_v = 27.9 \times P - 419 \text{ (K2)}$$



$$\Delta H_v = (2.37Cr - 3.02) \times T(20 + \log t) \times 10^{-3} - (34.8Cr - 31.5)$$

(430 < T [K] < 475 C, 5 < t [hour] < 1000, 13 < Cr [w/o] < 22)

Relationship yield, SP energy with ΔH_v



$$\Delta\sigma_{ys} [\text{MPa}] = 3.41 \times \Delta H_v, (0 < \Delta H_v < 120)$$

Summary

- Cathodic hydrogen charging considerably reduced the tensile ductility of ODS steels accompanied by the change of fracture mode from ductile shear rupture to intergranular and quasi-cleavage fracture.
- The critical hydrogen concentration required to induce brittle cracking in ODS steels is in the range of 10-12 wppm that is almost one order higher value than the 9Cr-2W RMS.

- High Cr ODS steels are experienced significant thermal aging embrittlement at temperatures from 430 to 475 C caused by the formation of Cr-rich coherent ferrite phase, α' phase.
- The hardening relationship at current test conditions can be obtained successfully as a function of Cr content and aging time.